Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

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The synthesis of a bimetallic nickel catalyst via the integration of different support materials comprising Al₂O₃, SiO₂, and dolomite was analysed and compared. The catalysts were synthesised using a microwave monomode reactor for 11 minutes at a temperature of 160 °C. The primary characterisation methods utilised for the investigation of reduction behaviour were temperature-programmed reduction (TPR), X-ray diffraction, and N₂ adsorption-desorption isotherm. The original structure of oxides Ni and Cr-Ni can be altered using silicon dioxide (SiO_2) , aluminium oxide (Al_2O_3) , and dolomite $(CaMg(CO_3)_2)$, which have all been subjected to research, indicating that there is a significant interaction between the catalyst and support. After the addition of the support, the distribution determined using Barrett-Joyner-Halenda (BJH) calculations indicates a broad pore size range that encompasses both meso (2-50 nm) and macro pore (>50 nm). In comparison to catalysts alone, oxides of Ni and Cr-Ni that contain SiO₂ have a greater surface area. Whereas SiO2 was found to effectively enhance the surface area of oxide Ni compared to Al_2O_3 and dolomite as supporting materials. The utilisation of SiO_2 on the Nibased catalysts caused the temperature of the TPR peak to decrease to 392 °C, compared to Al₂O₃ and dolomite. This may be attributed to two factors: weak interaction on SiO₂, the surface area, and porosity thus leading to easier diffusion of gas reductant, and accessibility of reactants during reduction thus enhancing the reduction of NiO species.

Keywords: Reduction behaviour; support catalyst; nickel oxide; bimetallic catalysts

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Significant modifications have been made in catalysis for renewable energy production, sensing, and electronics in the past decades due to developments in material science and engineering. Altering the properties of catalysts by changing their structure and composition provides researchers with the tools they need to better understand the fundamental properties of materials such as new spectroscopic, microscopic, and nanofabrication techniques [1]. This greatly improves multicomponent commercial catalysts in the field of catalysis.

Choosing a suitable support may be crucial for the catalyst's performance since it may imply a poor

or strong metal-support interaction that affects the adsorption ability and catalytic features [2]. Al₂O₃based support is often used in chemical reactions such as reforming due to its mechanical and chemical resistance [3]. Meanwhile, other support modifiers such as ZrO₂, SiO₂, MgO, La₂O₃, CeO₂, and TiO₂ also have been utilised, with the catalysts showing enhanced conversion, higher redox properties, high thermal stability, and resistance to sintering due to their favourable properties [2, 4, 5]. In addition to CaO, which is a naturally occurring metal oxide that is present around the planet, magnesium oxide (MgO) has also been thought of as an additive because it can be found in dolomite [6]. Due to its catalytic tar

Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

cracking and anti-sintering qualities, dolomite, a naturally occurring mineral of which primary component is $CaMg(CO_3)_2$, is desirable as a non-toxic base catalyst and CO_2 sorbent in thermochemical conversion processes such as biofuels [7].

Several catalyst preparation methods can be applied to obtain well-distributed active metal on support material that showed a significant impact on the catalyst's performance, such as impregnation, coprecipitation, sol-gel technique [8]. The impregnation is mostly utilised by researchers as it is the simplest approach for catalyst synthesis. However, the conventional wet impregnation method requires long hours of stirring [9–11] to achieve good dispersion of the catalyst on the support materials [12]. Besides that, a set temperature during stirring results in a low heating rate; the stirring condition also leads to greater heat loss. These phenomena can be prevented by adopting the microwave-assisted technique. One study used the microwave-assisted technique for developing bimetallic catalysts, proving their potential in providing homogenous internal and volumetric heating at high rates within a short time [13]. The radiation generated by microwave can provide better homogeneous heating, resulting in the nucleation of small particles to obtain nanoparticles [14]. Furthermore, it has been demonstrated that microwave procedures are more energy-efficient and environmentally benign than conventional processes, and that they have been effectively employed in both organic and inorganic syntheses [15]. Therefore, this work attempts to adopt microwave-assisted synthesis in catalyst preparation.

Active metals of nickel-based catalysts are widely recognised as commercial catalysts due to their low cost, ubiquitous availability, and excellent catalytic characteristics. However, due to metal agglomeration and carbon production, they are prone to deactivation. Previously, we had reported the chemical behaviour of mono and bimetallic Ni-based nanocatalyst via temperature programmed reduction (TPR) study [16, 17]. By suitably introducing support to the catalyst's surface, a catalytic synergistic effect between the active phases and the support may be achieved, increasing the reducibility of Ni and preventing sintering. Hence, in this work, we used microwaveassisted synthesis techniques to add support systems to the mono and bimetallic Ni-based catalysts. The synthesised catalysts were examined using powder X-ray diffraction (XRD), Brunauer-Emmett-Teller (BET) for surface area measurements, and temperature-programmed reduction (TPR). The effect of incorporating various support would be subsequently discussed.

EXPERIMENTAL

Sample Preparation

Mono and bi-metallic Ni-based with supported catalysts were prepared by microwave-assisted synthesis

using commercial SiO₂, Al₂O₃, and dolomite support with Ni(NO₃)₂.6H₂O and CrN₃O₉·9H₂O solutions, respectively. For microwave-assisted synthesis, nitrate precursors of Ni and Cr with 10% wt. (ratios 1:9) and SiO₂, Al₂O₃, and dolomite were placed in a sample tube containing 5 mL of ethanol liquid. The deionised water was measured using a new sample tube approximately half of the sample tube. Then, the deionised water was added to the mixture. The mixture was centrifuged at 600 rpm for 11 minutes at a temperature of 160 °C using a monomode microwave reactor. The mixture was dried in an oven overnight at 120 °C and the catalysts were ground to obtain a powder texture. Finally, it was calcined using a furnace at a temperature of 400 °C for 4 hours. All the prepared samples are denoted as NiO/SiO₂, NiO/Al₂O₃, NiO/Dolomite and CrNiO/SiO₂, CrNiO/ Al₂O₃, and CrNiO/Dolomite.

Characterisation Methods

The BET surface area and pore size distribution of the catalysts were obtained using a Micromeritics Tristar unit analyser with liquid nitrogen at 77.4 K as the adsorbate required for profiling the adsorption and desorption isotherms. A Bruker XRD-6000 diffractometer with Cu target K-a radiation source (40 kV, 30 mA) was used to analyse the crystalline phases of Ni0 and NiO in the sample. A scanning speed of 2° min⁻¹ was used over an angular range of $20^{\circ} < 2\theta < 80^{\circ}$ for the XRD characterisation. The estimation of the average Ni crystallite size of the catalysts was carried out upon calcination of fresh catalysts in an air atmosphere at 700 °C for 4 hours. A scanning speed of 0.2° min⁻¹ was used for 2 θ range between 40-50° and Scherrer's equation was used for the estimation of the NiO/Ni⁰ crystallite sizes.

The reducibility profiles of the catalysts were obtained from a CO-Temperature Programmed Reduction (CO-TPR) analysis which was conducted using an Autochem II Plus equipped with a thermal conductivity detector connected to a moisture trap. About 20 mL min⁻¹ of 40% CO/N₂ probe gas mixture was passed through ~50 mg of sample which was heated from ambient conditions to 900 °C at a ramping rate of 10 °C min⁻¹ to obtain the TPR profile.

RESULTS AND DISCUSSION

Catalysts Characterisation

Surface Study via N2 Ad-desorption Analysis

Figures 1a, 1b, and 1c show the isotherm curve for N_2 ad-desorption whereas Figure 1ai-bi-ci indicates the pore size distribution of the prepared catalysts. The surface area value, volume, and pore size calculated using the BJH method for as-prepared catalysts with fresh support materials and fresh catalysts are summarised in Table 1. The SiO₂ curve in Figure 1a is classified as typical of type IV with the hysteresis loop

Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

type H1 which indicates the capillary condensation effect that proves the existence of meso-pore composed of a uniform meso-pore range diameter [18, 19]. Type H1 is also classified with porous material consisting of agglomerate or compact uniform spherical arrangement with a narrow pore size distribution, in agreement with Figure 1ai [20]. Meanwhile, for Al₂O₃ (Figure 1b) and dolomite (Figure 1c), the isotherms curve is typical of type III with loop H3 exhibiting a porous existence or consisting of macro-pores, plate-like rigid aggregates, and a network of pores comprising macro-pores that are not completely filled with burrow condensation [19]. Furthermore, the Al_2O_3 and Dolomite samples are composed of a combination of meso and macro pores, in agreement with Figure 1bi and ci. As reported in the catalyst study, mesopore structure can provide good benefits and resistance to Ni sintering [21].

After the addition of a support system (SiO_2) into the mono and bimetallic catalysts, the isotherms remain and are classified as type IV with loop H1 which implies the formation of high mesopore [22]. Meanwhile, catalysts with Al₂O₃ supporter show typical physical adsorption of type IV but with the shape of a hysteresis loop H3. Occasionally, 10CrNiO-dolomite shows type V properties except for 10Ni-Dolomite (Figure 1c) which exhibits the properties of type III, indicating the presence of micropore structures. All samples exhibit a loop shape, indicating the presence of a mesopore structure [23]. This is in contrast to the fresh NiO and Cr-Ni without support which show typical properties of type III and typical IV hysteresis loop H3 (not shown here). This suggests that the presence of SiO₂ and dolomite materials on the mono and bimetallic catalysts completely dominates the initial catalyst structure except for Al_2O_3 .

SiO₂ material shows a high surface area of 115.00 m² g⁻¹, compared to Al₂O₃ and Dolomite with values of 0.66 and 0.81 m² g⁻¹, respectively. The surface area of Al₂O₃ exhibits α-Al₂O₃ properties with a low surface area or better known as α -Al₂O₃ (<10 m² g⁻¹) [24]. It is well known that the surface area of α -Al₂O₃ is lower than that of γ -Al₂O₃ [25]. Alumina can exist in various phases such as Alpha (α) and Gamma (γ)-alumina phases. Indeed, α -Al₂O₃ has thermodynamic stability and some properties such as thermal stability, "high hardness", and stability of chemical properties. Meanwhile, the γ -Al₂O₃ is an unstable phase as it can easily switch phases at a low temperature of 600-700 °C and requires a high temperature of 950 °C for phase conversion to α-Al₂O₃ [26].



Figure 1. (a, b, c) N₂ adsorption-desorption analysis and (ai, bi, ci) BJH pore size distribution plots as-prepared catalysts.

Catalysts	Surface Area $(m^2/g)^a$	average pore width (nm) ^b	Pore Volume (cm ³ /g) ^b	Crystallite size (nm) ^c
NiO	2.63	17.30	0.004	38.90
SiO ₂	115.00	19.77	0.744	48.80
Al ₂ O ₃	0.66	13.11	0.001	78.80
Dolomite	0.81	17.45	0.002	1.00
10NiO-SiO ₂	323.00	6.46	0.717	12.20
10CrNiO-SiO ₂	371.00	6.43	0.687	7.00
10NiO-Al ₂ O ₃	5.90	6.53	0.007	19.00
10CrNiO-Al ₂ O ₃	7.62	19.27	0.023	27.00
10NiO-Dolomite	3.37	19.47	0.006	21.50
10CrNiO-Dolomite	13.25	7.36	0.018	10.20

 Table 1. BET surface area, pore volume, and average pore width as-prepared catalysts calculated from Figure 1 and Figure 2.

^aBET

^bBJH

^cScherrer equation XRD

The structural properties of supported Ni and Cr-Ni oxide with the addition of support materials as shown in Table 1 demonstrates that the catalyst series with SiO₂ has a high surface area with a range of 323.00-371.00 m² g⁻¹ compared to α-Al₂O₃ supported catalysts (5.90-7.62 m² g⁻¹) and dolomite (3.37-13.25 m^2 g⁻¹). The presence of SiO₂ increases the surface area of Ni and Cr-Ni oxide catalysts compared to unsupported ones. However, it can be seen that the increase in surface area of the Ni oxide catalyst is only effective by using SiO₂ compared to Al₂O₃ and dolomite. The increase in catalyst surface area can be attributed to the formation of new external pores on the surface of NiO and the existence of mesopore formed after the addition of SiO₂ support, which then forms a new layer on the external surface of the catalyst [27]. The use of SiO₂ material can increase the porosity of the catalyst and can form a network of interconnected pores in the Ni matrix [28]. In addition, it has been proven that the addition of SiO₂ can help spread Ni particles more uniformly and prevent agglomeration as evidenced by the small crystal size (Table 1). In general, the formation of small crystal sizes will lead to an increase in the surface area of the catalyst after calcination [29].

Figure 2 shows an analysis of XRD diffractogram phase changes prior to the addition of a supporting system into the mono- and bi-metallic metal based on Ni catalysts. Generally, by comparing the mono and bimetal supported catalysts with NiO alone, the surface area for the catalysts supported is inversely proportional to the NiO crystallite size. The crystallite size for fresh NiO and supported were measured using a lattice plan with the highest intense peak at [200], [102], [113], and [006], respectively. The peaks that appear at 20:37.3°, 43.3°, 62.9°, 75.4°, and 79.7° can be attributed to the existence of the NiO phase (ICDD 01-078-0643) while the wide peak at an angle of $\sim 22^{\circ}$ can be described as the SiO₂ phase (ICDD 00-033-1161). The peak intensity increases for 10NiO-SiO₂ and there is a slight displacement of the 2θ angle of the NiO phase to a higher position which indicates a small contraction in the unit cell parameters during the catalyst preparation [30]. This is because nickel crystals have entered the SiO₂ pores and in turn change the lattice of the SiO₂ structure [31]. A lower intensity is observed at 20:22° for 10NiO-SiO₂ compared to 10CrNiO-SiO₂, which shows that the SiO₂ crystal structure is disturbed due to the incorporation of Ni ions into the lattice structure of the supports, leading to a decrease in crystal coherence. Furthermore, for bimetallic catalysts, the XRD spectrum showed a wide SiO₂ signal peak at 2θ :~22° with low/weak NiO crystal peaks visible at lattice plane positions [111], [200], and [220] and a new appearance peak at 20: 34.7°, which correspond to NiCrO₄ (ICDD 01-073-1046). This reveals that the addition of a second metal altered the electronic structure of the Ni species into a nano size. Table 1 shows that the addition of Cr decreased the crystal size of the Ni species (7.0 nm) compared to 10NiO-SiO₂ (12.2 nm). The decrease in crystal size may result from the addition of Cr and Ni, which can cause changes in crystal structure or growth mechanism with SiO₂ material. The presence of a second metal can lead to the formation of different phases of crystals or changes in the nucleation and growth rate of the crystals, which will affect the size of the final crystal after calcination. This is significant with the decrease in crystallite size of the Ni species in the CrNiO-SiO₂ sample. The nickel oxide phase is located in the meso pores and this is in line with the XRD patterns (Figure 2a), which shows the high intensity of the Ni oxide species (large crystal size) of sample NiO-SiO₂. The high intensity peaks of NiO-SiO₂ are also due to

Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

the concentrated dispersion of X-rays to the highly crystalline NiO species versus SiO₂ which in turn produces low SiO₂ intensity peaks on the XRD diffractograms [32]. As a result, Ni species particles scattered well above SiO₂ after going through the drying process upon the addition of the Cr promoter [33], which is in agreement with the increase in surface area as shown in Table 1. Figure 2b shows that Al₂O₃ peaks are present at positions 20: 25.7, 35.2, 37.9, 43.4, 52.6, 57.6, 61.4, 66.7, 68.3, and 77.1° with ICDD reference numbers 00-010-0173, respectively [26]. Meanwhile, the angle at position 20: 37.4°, 43.4°, 63.0°, 75.7°, and 79.6° with lattice [111], [200], [220], [311], and [222] corresponds to the NiO species (ICDD 01-078-0643) while a new peak crystal at 20: 35.9° corresponds to the formation of alloy NiCrO₄ with ICDD 00-073-1046. The crystal size of NiO is certainly measured based on the highest peak angled at 20: 43.4°. However, the peak is overlapped with the peak of NiO and Al₂O₃; therefore, it is likely that the oxide phases of Ni and Al have formed a new species of NiAl₂O₄ at peaks 20: 37°, 42° and 63° (ICDD 00-001-1299), which is in agreement with previous research [2]. Thus, the crystallite size for Ni species for the prepared catalyst supported with Al₂O₃ is determined by reference to the lattice plane of nickel oxide [111] at an angle of 20: 37.3°. The addition of Al₂O₃ support increases the size of the nickel crystals compared to catalysts supported by SiO₂ (Table 1).



51 Salma Samidin, Azizul Hakim Lahuri, Wan Nor Roslam Wan Isahak, Khairul Naim Ahmad, Muhammad Rahimi Yusop and Norliza Dzakaria



Figure 2. XRD diffractogram of mono- and bi-metallic Ni-based catalyst supported by (a) SiO₂, (b) Al₂O₃, and (c) Dolomite.

By comparing the catalysts supported by Al₂O₃ with catalysts supported by SiO₂, the surface area of catalysts supported by Al₂O₃ is inversely proportional to the crystallite size of the NiO. This might correspond to the addition of a metallic charge to Al₂O₃ which causes particles to form during solid drying, limited to producing new pores. Figure 2c shows the presence of dolomite (CaMg(CO₃)₂) phase peaks at position 2θ : 22.1-75.4° (ICDD 00-075-1711). The addition of dolomite into mono- and bi-metal oxide shows the decrease in crystal size for the Ni species (Table 1). The addition of Cr shows an alloy peak at $20:24.5^{\circ}$, which corresponds to NiCrO₄ (ICDD 00-021-0596). Overall, the XRD study shows the decrease in crystal size for the NiO species in mono- and bi-metallic for all the prepared catalysts compared to unsupported NiO. The formation of small crystallite is due to the interaction between NiO and support materials (nucleation sites) which alter and inhibit the growth of larger crystallite [34].

Reducibility Study

Temperature Programmed Reduction (TPR), as shown in Figure 3, was used to examine the reducibility of oxide species and their interactions with various support systems. We conducted a comparative analysis using unsupported mono- and bi-metallic Nibased catalysts, referring to our previous studies, to understand the effect of various support systems on NiO reduction [17]. Three steps of NiO reduction stages I, II, and III were recorded at the temperature range of 400-700 °C [16]. Different NiO species were found in SiO₂, Al₂O₃, and dolomite when the TPR curve was deconvolved and fitted. Figure 3a shows four different reduction peaks. The initial and second peaks for NiO-SiO₂, NiO-Al₂O₃, and NiO-Dolomite took place at 392, 404, and 478 °C (red line) and 436, 460 and 539 °C (green line), indicating the partial reduction of NiO species at surface while Bourdouard reaction (2CO (g) \rightarrow CO₂ (g) + C (s)) occurred during the reduction process which in agreement with a previous study [35]. According to earlier studies [24, 36], this peak represents a reduction of weaklyattached NiO with relatively limited dispersion on the support system. It is interesting to note that the dolomite support induced a rise in the NiO reduction temperature to 478 and 539 °C, which may be the result of a moderate interaction between bulk oxide species on the dolomite crystal surface [21]. Convolution peaks also occurred at 506 °C (NiO-SiO₂), 502 °C (NiO-Al₂O₃), and 645 °C (NiO-dolomite) (blue line) and at 584 °C (NiO-SiO₂), 730 °C (NiO-Al₂O₃), and 790 °C (NiO-dolomite) (pink line), which are related to the reduction of spinel type NiO species and solid solution (i.e NiO-CaO, NiO-MgO). It is noteworthy that the peak region of the pink curve (NiO-SiO₂) shows a high concentration of NiO actively interacting with SiO₂. According to this pattern, SiO₂ increased the reducibility of NiO species, probably as a result of better metal dispersion, as corroborated by the XRD diffractogram data (Figure 2a). The well-dispersed NiO particles may have formed more easily due to SiO₂'s greater surface area, increasing the amount of active sites available for reduction [37].

With regards to bimetallic CrNi oxide, Figure 3b illustrates its reducibility behaviour as supported by SiO₂, Al₂O₃, and dolomite. Shoulder peaks at 203, 268, and 345 °C for CrNiO-SiO₂, CrNiO-Al₂O₃, and CrNiO-Dolomite, respectively, are associated with the surface-level conversion of Cr^{6+} to Cr^{3+} [17]. Contrary to the supported monometallic, the addition of support systems into the bimetallic catalyst led to a shift

52 Salma Samidin, Azizul Hakim Lahuri, Wan Nor Roslam Wan Isahak, Khairul Naim Ahmad, Muhammad Rahimi Yusop and Norliza Dzakaria

Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

towards lower temperatures. This change may be attributable to the synergistic actions of Cr and Ni, which improve the overall reducibility qualities. It is interesting to note that the CrNiO-Al₂O₃ curve (blue line) showed a broad and intense peak, indicating higher CO dissociation at 531 °C. This result suggests a larger concentration of species that can be reduced and effectively reduced [19]. Three different Ni species (α, β, γ) have been discovered in NiO/Al₂O₃ catalysts in earlier investigations. The α -type species have poor metal-support interactions and are active at low temperatures, whereas the β -type interacts more strongly with the support and promotes region reduced at moderate temperatures. NiAl₂O₄, which represents the γ -type, can only be reduced at high temperatures (750-900 °C) [21]. The expansion of the TPR peak for Ni and Cr-NiO/dolomite between 500 and 600 °C may be due to the coexistence of two reduction stages:

homogenous NiO-MgO solid solution and free-state NiO (poor contact with support) [38]. As the temperature rises, the peak gradually goes down, indicating that there is a uniform NiO-MgO solid solution, which is one of the hardest phases to reduce. Dolomite introduces a number of oxide-support metal interaction phases, including solid solutions (NiO-MgO) and spinel NiO (NiCr₂O₄), which are difficult to distinguish [21]. Compared to SiO₂ and Al₂O₃, the NiO reduction for CrNiO/dolomite was more evident at higher temperatures for stage I, as previously reported [39]. The reduction temperature further shifted to 629 °C with three overlap deconvulated peaks at 345, 492, and 602 °C due to the interaction between Ni and the components of dolomite, which creates the stable nickel phase NiMgO₂ [40]. Meanwhile, stage II for NiO located at 645 and 790 °C and Cr-Ni samples at 776, 689, 808, and 845 °C indicated the reduction of strongly bound Ni²⁺ ions



Figure 3. The deconvolution integration curve of TPR profile for (a) mono- and (b) bi-metallic Ni-based catalysts supported by SiO₂, Al₂O₃, and Dolomite

and oxide clusters inside the dolomite pores [41, 42]. Stage II for metal oxide supported by SiO₂ corresponds to NiO reduction with significant interaction with SiO₂ [39]. Additionally, the existence of crystals within the supporting pores, which is a result of the metal extraction process during calcination, may be responsible for the broad peak of CrNiO-Al₂O₃ reaching close to 900 °C [43]. This finding is consistent with the BET analysis, which shows that CrNiO-Al₂O₃ has greater pore volume and diameter.

CONCLUSION

The TPR results shed light on the reducibility behaviour of NiO and bimetallic CrNi oxide supported by SiO₂, Al₂O₃, and dolomite. The reduction characteristics and active species interactions are heavily influenced by the support materials. It is noteworthy that the CrNiO-SiO₂ exhibited the lowest reduction temperature at 392 °C compared to CrNiO-Al₂O₃ and CrNiO-Dolomite due to weak interaction between CrNiO and SiO₂. The findings are critical for catalyst design and optimisation, allowing for application-specific catalytic performance. Future studies might look into enhanced characterisation techniques to better understand the relationships and processes behind the observed patterns.

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Surface and Textural Analyses for Bimetallic Nickel-Based Catalyst onto Various Support using Microwave Assisted Synthesis

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